

LAMINATED STIFFNESS (HW2)

HM CARBON

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AEROSPACE MATERIALS

(14:650:449:90)



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PROBLEM STATEMENT:

ANALYZE THE STIFFNESS AND DEFORMATION RESPONSE OF A SYMMETRIC LAMINATE COMPOSED OF HIGH MODULUS CARBON PLYS.

LAMINATE STACKING SEQUENCE:

[+ θ / - θ / +30 / -30]_S

Objectives:

- Compute laminate engineering constants
- Determine laminate strains under applied stress
- Determine ply stresses in material coordinates
- Study how laminate behavior varies with orientation angle θ



INPUT DATA - HM CARBON

Table 6: Elastic Properties of Unidirectional Polymer Matrix Composites and 7075-T6 Aluminum

Material Properties:	Value:
E11	225 GPa
E22	7 GPa
G12	3.9 GPa
ν_{12}	0.25

Applied Stress State:	Value:
σ_{xx}	1799 MPa
σ_{yy}	282 MPa
τ_{xy}	672 MPa

Ply Thickness: $h = 0.0001$ m

METHODS OF ANALYSIS

Using the matrix definitions shown and the HM Carbon material properties, laminate strains and ply stresses were computed for different fiber orientation angles. The effective engineering constants were then evaluated as functions of the orientation angle θ .

Material Compliance Matrix:

$$[S_1] = \begin{bmatrix} 1/E_{11} & -\nu_{12}/E_{11} & 0 \\ -\nu_{12}/E_{11} & 1/E_{22} & 0 \\ 0 & 0 & 1/G_{12} \end{bmatrix}$$

Material Stiffness Matrix:

$$[C_1] = [S_1]^{-1}$$

Transformation matrices:

$$m = \cos\theta \quad n = \sin\theta$$
$$[T_1] = \begin{bmatrix} m^2 & n^2 & 2mn \\ n^2 & m^2 & -2mn \\ -mn & mn & m^2 - n^2 \end{bmatrix} \quad [T_2] = \begin{bmatrix} m^2 & n^2 & mn \\ n^2 & m^2 & -mn \\ -2mn & 2mn & m^2 - n^2 \end{bmatrix}$$

Calculation Steps:

$$\begin{aligned} \{\sigma_1\} &= [T_1] \{\sigma_2\} & [S_2] &= [T_2]^{-1} [S_1] [T_1] \\ \{\epsilon_1\} &= [S_1] \{\sigma_1\} & \{\epsilon_2\} &= [S_2] \{\sigma_2\} \\ \{\epsilon_2\} &= [T_2]^{-1} \{\epsilon_1\} & \{\epsilon_1\} &= [T_1] \{\epsilon_2\} \end{aligned}$$

Matrix definitions and calculation steps used in the laminate analysis

RESULTS OF ANALYSIS

Ply Stresses:	θ :	$(-\theta)$:	$(+30)$:	(-30) :
ϵ_{11} :	0.012403	-0.0009	0.020134	0.008121
ϵ_{22} :	-0.02264	-0.00933	-0.03037	-0.01836
γ_{12} :	-0.06994	0.077773	-0.05974	0.07361
σ_{11} :	2756.348	-220.335	4485.783	1798.568
σ_{22} :	-137.055	-67.0542	-177.724	-114.531
τ_{12} :	-272.764	303.314	-232.979	287.079

Engineering Const:	
E_{xx} :	40396.202
E_{yy} :	10226.567
G_{xy} :	48443.36
ν_{xy} :	1.5985373
ν_{yx} :	0.4046803

Laminate Strains:	
ϵ_{xx} :	0.033375
ϵ_{yy} :	-0.04361
γ_{xy} :	0.013872

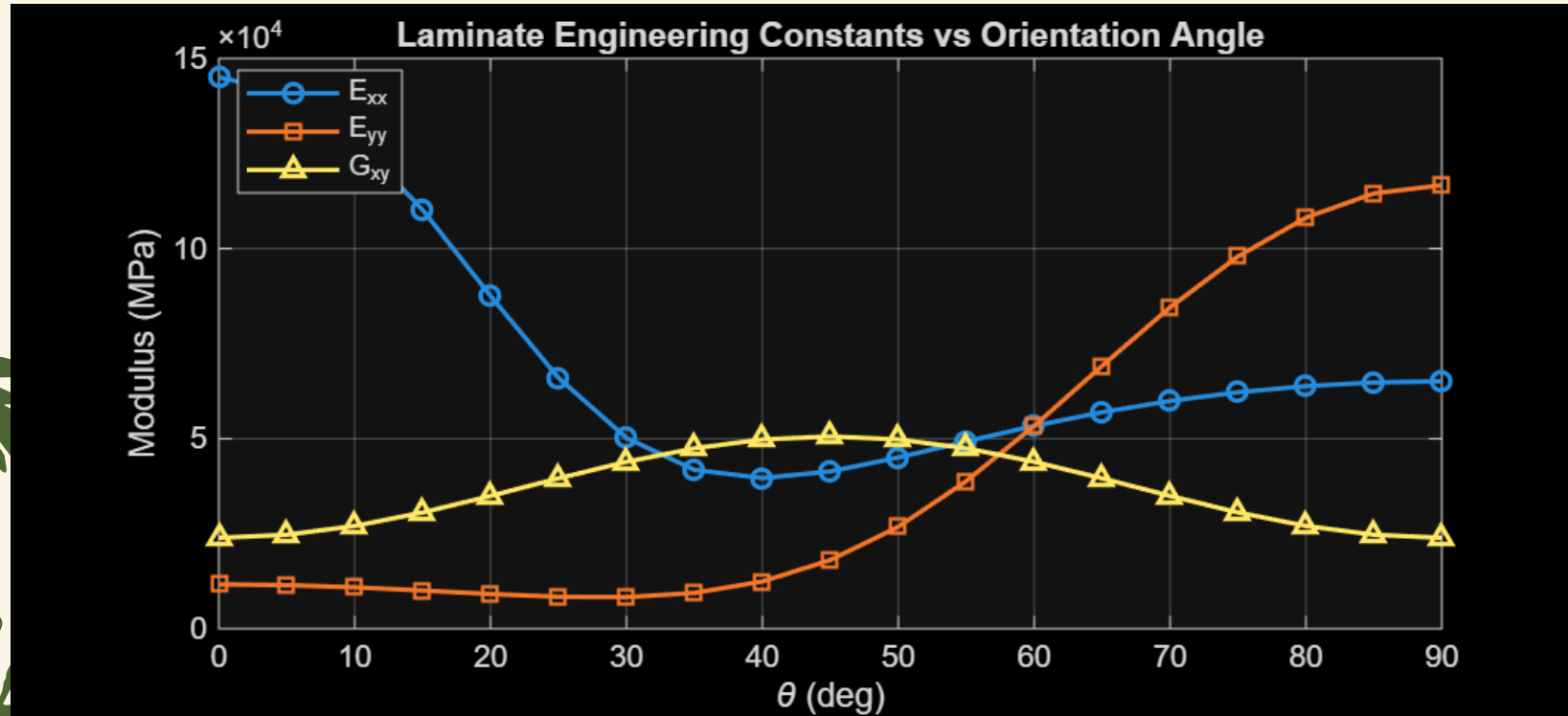


RESULTS OF ANALYSIS CONT.

Using the HM Carbon material properties and the laminate stacking sequence $[+\theta/-\theta/+30/-30]_s$, the engineering constants, laminate strains, and ply stresses were calculated for the assigned angle $\theta = 36.8^\circ$. The results show that the laminate is much stiffer in the x -direction than in the y -direction, since E_{xx} is significantly larger than E_{yy} . The applied stress causes a positive strain in the x -direction and a negative strain in the y -direction, along with a shear strain due to the applied shear stress. The ply stress results also show that each ply carries different stress values depending on its orientation. This highlights how fiber direction plays a major role in how the laminate responds to loading.

RESULTS OF ANALYSIS CONT.

E_{xx} MAX = 145075.21 MPA AT θ = 0.0 DEG
E_{xx} MIN = 39611.22 MPA AT θ = 40.0 DEG
E_{yy} MAX = 116661.36 MPA AT θ = 90.0 DEG
E_{yy} MIN = 8290.89 MPA AT θ = 30.0 DEG
G_{xy} MAX = 50569.26 MPA AT θ = 45.0 DEG
G_{xy} MIN = 23901.11 MPA AT θ = 0.0 DEG



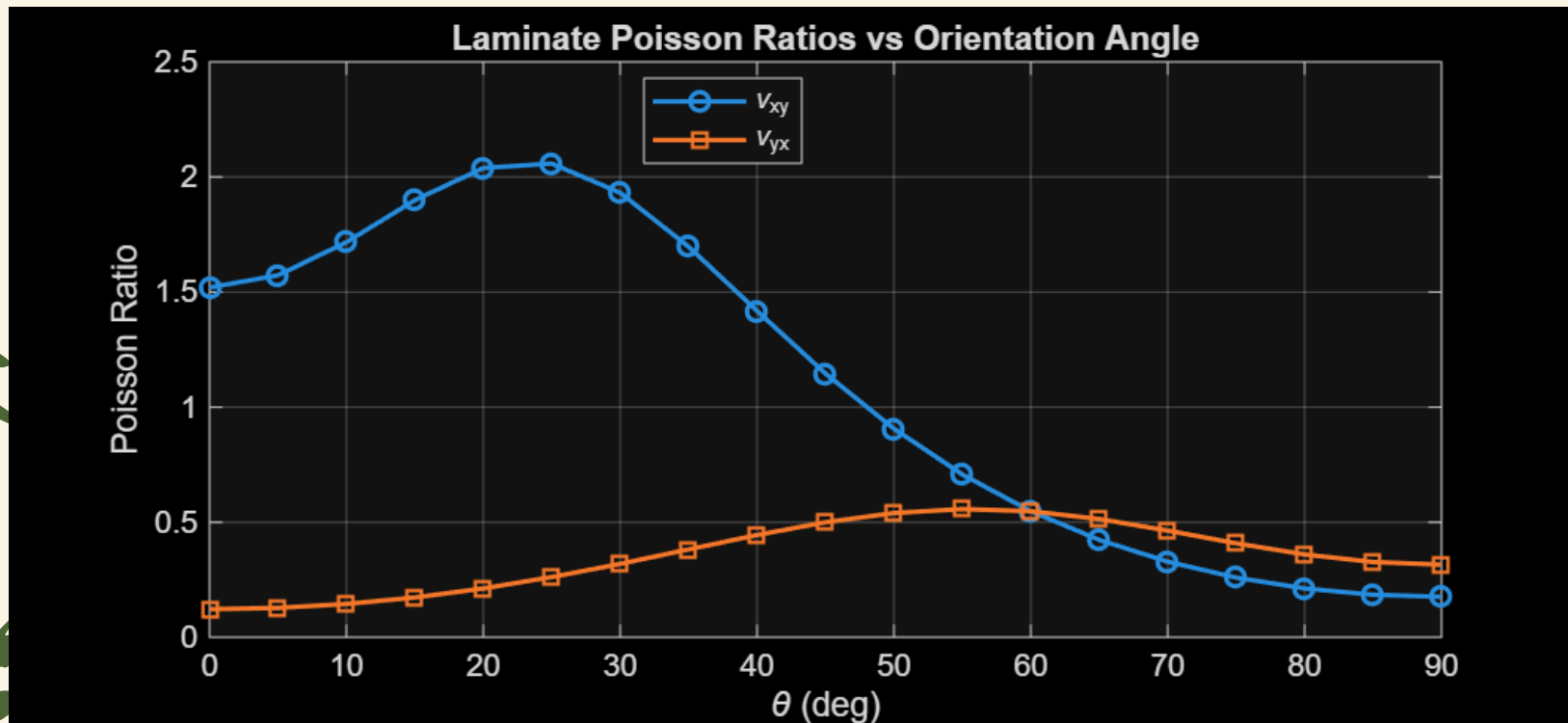
- E_{xx} decreases as θ increases, showing that stiffness in the loading direction drops as fibers rotate away from the x-direction.

- E_{yy} increases toward 90° , indicating that stiffness shifts toward the transverse direction.

- G_{xy} reaches its maximum near 45° , where the laminate experiences the greatest shear interaction.

RESULTS OF ANALYSIS CONT.

VXY MAX = 2.0583 AT $\theta = 25.0$ DEG
VXY MIN = 0.1761 AT $\theta = 90.0$ DEG
VYX MAX = 0.5572 AT $\theta = 55.0$ DEG
VYX MIN = 0.1221 AT $\theta = 0.0$ DEG



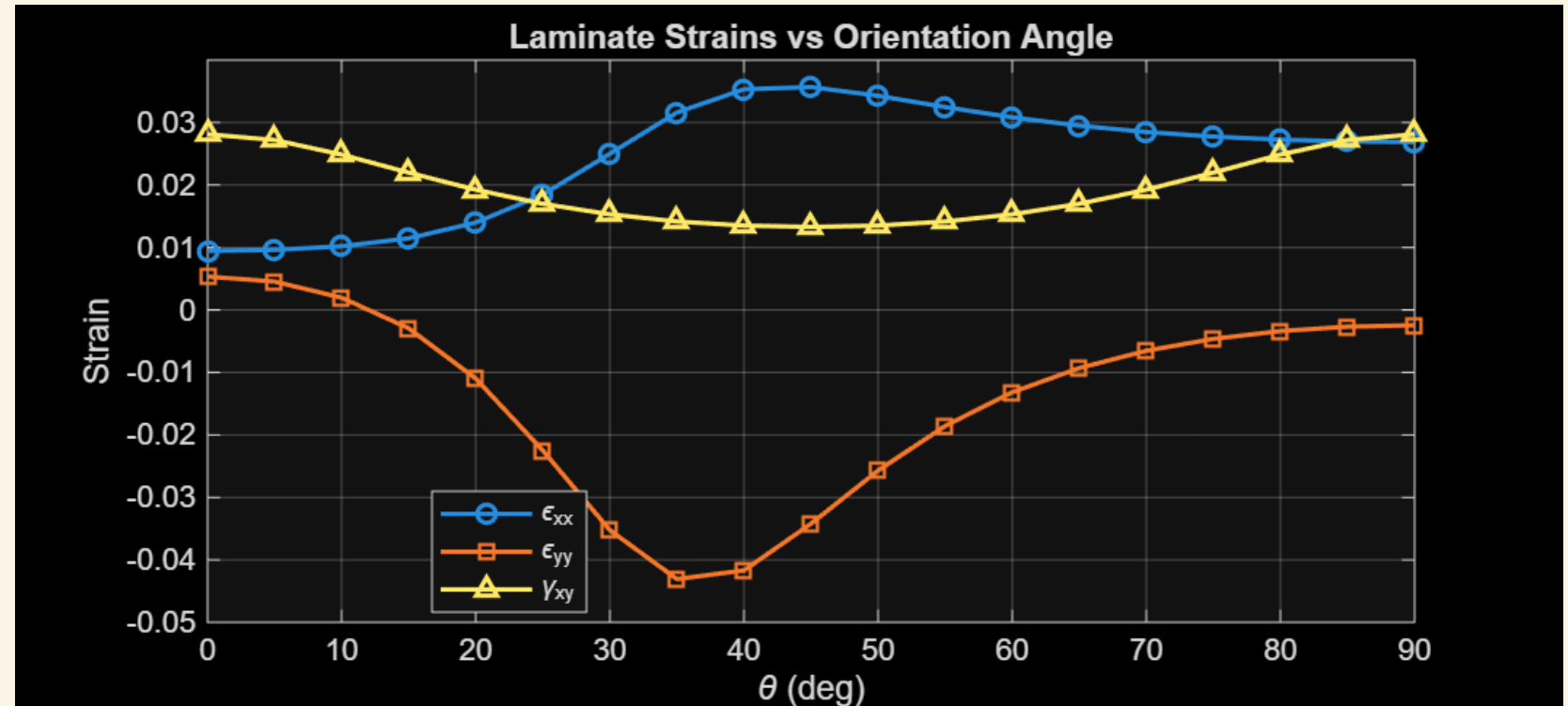
- The Poisson ratios vary strongly with fiber orientation, showing the anisotropic behavior of the laminate.
- v_{xy} reaches its maximum near 25° , while v_{yx} peaks near 55° , indicating different coupling effects between the x and y directions.

RESULTS OF ANALYSIS CONT.

• The laminate strains change with fiber orientation because the effective stiffness of the laminate varies with θ .

• ϵ_{xx} increases as the laminate becomes less stiff in the x-direction, while ϵ_{yy} becomes more negative due to the applied loading.

EPS_XX MAX = 0.035676 AT θ = 45.0 DEG
EPS_XX MIN = 0.009447 AT θ = 0.0 DEG
EPS_YY MAX = 0.005343 AT θ = 0.0 DEG
EPS_YY MIN = -0.043071 AT θ = 35.0 DEG
GAMMA_XY MAX = 0.028116 AT θ = 0.0 DEG
GAMMA_XY MIN = 0.013289 AT θ = 45.0 DEG

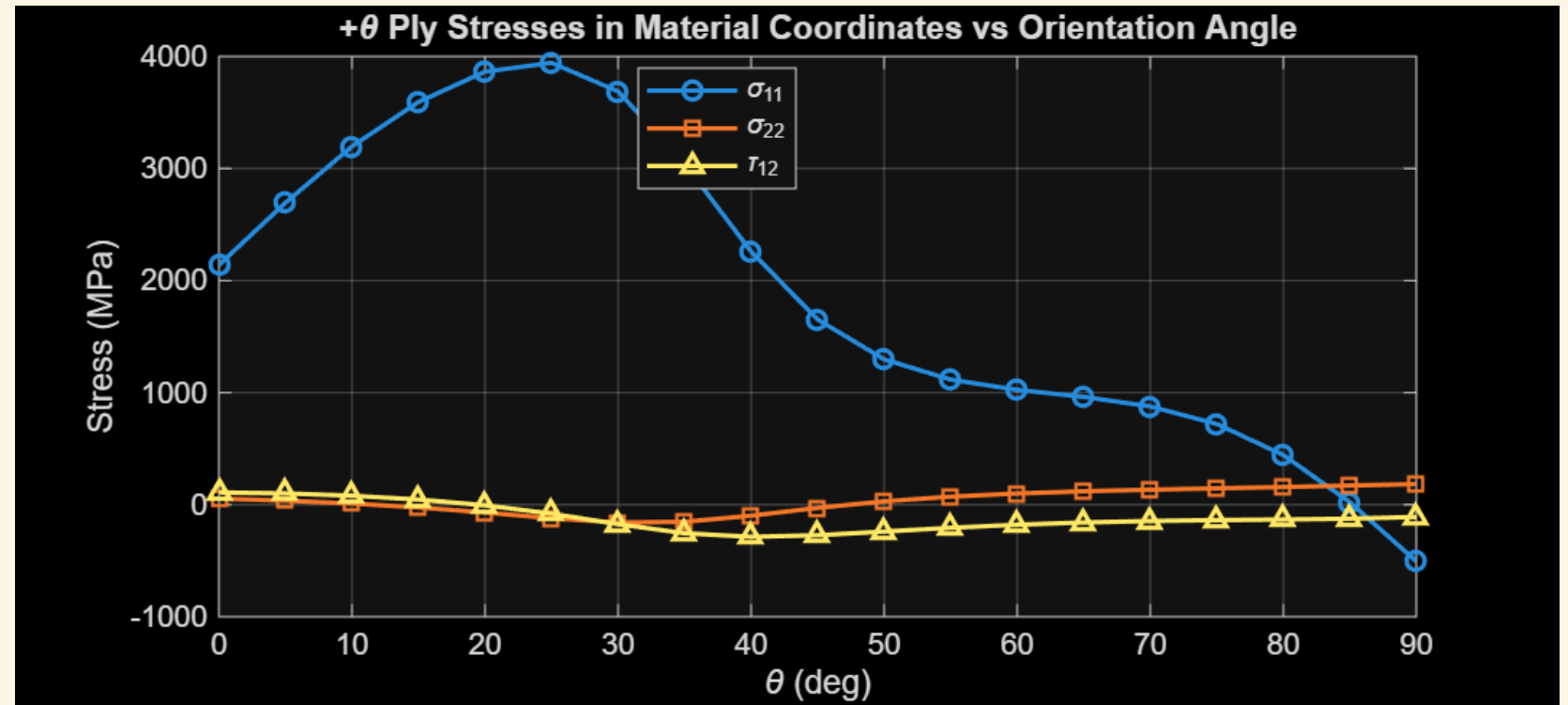


RESULTS OF ANALYSIS CONT.

- Ply stresses vary significantly with fiber orientation because each ply carries load differently depending on its alignment with the applied stress.

- The longitudinal stress σ_{11} is highest when the fibers are closer to the loading direction.

SIGMA11 MAX = 3942.94 MPA AT θ = 25.0 DEG
SIGMA11 MIN = -505.23 MPA AT θ = 90.0 DEG
SIGMA22 MAX = 184.20 MPA AT θ = 90.0 DEG
SIGMA22 MIN = -158.98 MPA AT θ = 30.0 DEG
TAU12 MAX = 109.65 MPA AT θ = 0.0 DEG
TAU12 MIN = -286.64 MPA AT θ = 40.0 DEG




OVERALL DISCUSSION:

The results show that the mechanical behavior of the laminate depends strongly on the fiber orientation angle θ . As the fiber direction changes, the stiffness properties E_{xx} , E_{yy} , and G_{xy} vary significantly. When the fibers are aligned closer to the x -direction, the laminate shows higher stiffness in that direction, which results in lower strain under the applied loading. As the fibers rotate away from the loading direction, the laminate becomes less stiff in the x -direction and the strain increases. The shear modulus reaches its maximum near 45° , which indicates that shear resistance is strongest when fibers are oriented diagonally relative to the loading direction. The Poisson ratios also change with fiber orientation, showing the anisotropic nature of composite laminates. This means the deformation in one direction is strongly affected by the fiber alignment.

The laminate strains follow the same trends as the stiffness values. As stiffness decreases in a direction, the strain in that direction increases under the same applied stress. The ply stress results show that individual plies experience very different stress levels depending on their orientation. Plies aligned closer to the loading direction carry higher longitudinal stresses, while off-axis plies experience larger shear stresses. Overall, these results highlight how fiber orientation plays a major role in determining the stiffness, strain response, and stress distribution of a composite laminate. This is why fiber orientation is a key design parameter when engineering composite structures.

CONCLUSION



THIS ANALYSIS APPLIED CLASSICAL LAMINATE THEORY TO EVALUATE THE STIFFNESS, STRAIN RESPONSE, AND PLY STRESSES OF A SYMMETRIC HM CARBON LAMINATE. THE RESULTS SHOW THAT LAMINATE MECHANICAL PROPERTIES ARE HIGHLY DEPENDENT ON THE FIBER ORIENTATION ANGLE θ , DEMONSTRATING THE ANISOTROPIC BEHAVIOR OF COMPOSITE MATERIALS. AS THE FIBER ORIENTATION CHANGES, THE EFFECTIVE STIFFNESS SHIFTS BETWEEN THE LONGITUDINAL AND TRANSVERSE DIRECTIONS, WHICH DIRECTLY AFFECTS THE STRAIN RESPONSE OF THE LAMINATE UNDER THE APPLIED LOADING. THE ANALYSIS ALSO SHOWED THAT STRESSES ARE DISTRIBUTED DIFFERENTLY AMONG THE PLYS DEPENDING ON THEIR ORIENTATION, MEANING THAT SOME PLYS CARRY HIGHER LOADS WHILE OTHERS EXPERIENCE GREATER SHEAR EFFECTS. THESE RESULTS HIGHLIGHT HOW FIBER ORIENTATION CAN BE USED AS A DESIGN PARAMETER TO TAILOR THE MECHANICAL PERFORMANCE OF COMPOSITE LAMINATES FOR SPECIFIC LOADING CONDITIONS. OVERALL, THIS STUDY DEMONSTRATES HOW LAMINATE THEORY PROVIDES A USEFUL FRAMEWORK FOR PREDICTING THE BEHAVIOR OF COMPOSITE STRUCTURES AND GUIDING ENGINEERING DESIGN DECISIONS.

EXCEL AND MATLAB FILE PROVIDED AS A SEPARATE ATTACHMENT.

```

/MATLAB Drive/untitled333.m
1 clear; clc; close all;
2
3 %% =====
4 % INPUT DATA
5 % =====
6 E11 = 225000; % MPa
7 E22 = 7000; % MPa
8 v12 = 0.25; % unitless
9 G12 = 3900; % MPa
10
11 sigma_x = [1799; 282; 672]; % [sigma_xx; sigma_yy; tau_xy] in MPa
12 h = 0.0001; % ply thickness, m
13 alpha = 30; % fixed angle in lay
14
15 theta_vals = 0:5:90; % use 5 deg incremen
16
17 %% =====
18 % MATERIAL MATRICES
19 % =====
20 S1 = [ 1/E11 -v12/E11 0;
21 -v12/E11 1/E22 0;
22 0 0 1/G12 ];
23
24 C1 = inv(S1);
25
26 %% =====
27 % STORAGE ARRAYS
28 % =====
29 nPts = length(theta_vals);
30
31 Exx = zeros(nPts,1);
32 Eyy = zeros(nPts,1);
33 Gxy = zeros(nPts,1);
34 vxy = zeros(nPts,1);
35
36
37
38
39

```

	A	B	C	D	E	F	G	H	I	J
1	Material:	HM Carbon							HW2 - Laminated Stiffness	Aerospace Materials (14:650:449:90)
2	θ:	36.8		E11 (Gpa):	225	--> Mpa:	225000			By Avanthika Vuppala
3	α:	30		E22 (Gpa):	7	--> Mpa:	7000			
4	σ _{xx} (MPa):	1799		v12:	0.25					
5	σ _{yy} (MPa):	282		G12 (Gpa):	3.9	--> Mpa:	3900			
6	τ _{xy} (MPa):	672								
7	ply thickness h:	0.0001								
8										
9	[S1] Compliance Matrix:									
10		4.44444E-06	-1.11111E-06	0						
11		-1.11111E-06	0.000142857	0	<-- Tells you how the material strains when stress is applied in its own fiber directions					
12		0	0	0.00026						
13										
14	[C1] Stiffness Matrix:									
15		225438.3524	1753.409407	0						
16		1753.409407	7013.637629	0	<-- Inverse of [S1]					
17		0	0	3900						
18										
19										
20	cosθ:	0.800731371								
21	sinθ:	0.599023599								
22	cosα:	0.866025404								
23	sinα:	0.5								
24										
25										
26	[T1] Transformation Matrix:					[T2] Transformation Matrix:				
27		0.641170728	0.358829272	0.95931	0.641170728	0.3588293	0.47966			
28		0.358829272	0.641170728	-0.95931	0.358829272	0.6411707	-0.47966			
29		-0.479656987	0.479656987	0.28234	-0.959313975	0.959314	0.28234			
30										
31	[Cx] ^(θ) stiffness matrix of a ply rotated by θ:									
32		97976.67144	50837.90904	66830.8						
33		50837.90904	36306.31927	37938.1						
34		66830.82416	37938.11645	52984.5						
35										
36										
37	cos(-θ):	0.800731371								
38	sin(-θ):	-0.599023599								
39	[T1] for (-θ):					[T2] for (-θ):				